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## Electronic, thermodynamic, and effect of pressure on mechanical properties in CoTiSb and CoTiBi semi-Heuslers

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### **ABSTRACT**

First principles technique based on density functional theory (DFT) is employed to study the structural, phonon, electronic, thermodynamic, and mechanical properties of CoTiSb and CoTiBi semi-Heuslers. The optimized structural parameters including lattice constant, bulk modulus, and formation energy were determined in three structural phase. The results obtained were found to be in agreement with available experimental and other theoretical results. CoTiSb and CoTiBi were found to be brittle, mechanically, dynamically, and thermodynamically stable, and exhibit an indirect band gap semiconductor property as revealed by the bandstructure, total and projected density of states. The effect of pressure on the elastic, and mechanical behaviors were investigated in a range of 0 - 30 GPa. Thermodynamic properties of CoTiSb and CoTiBi were calculated from 0 – 800 K. At room temperature, the Debye heat capacity  $(C_v)$  was found to be 68.6 and 70.43 J/K/mol for CoTiSb and CoTiBi. At low temperature,  $C_v$ , obeys the Debye power-law ( $C_v \propto T^3$ ), and at eleviated temperature,  $C_{v}$ , agrees with the Dulong-Pitit law, and converged at a value of 73.94 and 74.22 J/K/mol for CoTiSb and CoTiBi, respectively. For the first time results have been obtained for CoTiBi semi-Heusler. These results may serve as reference to further studies.

### 1. INTRODUCTION

Full Heusler (FH) and semi-Heusler (SH) compounds have attracted worldwide interest due to their promising applications optoelectronics, spintronics, thermoelectrics (Benzoudji et al., 2019; Casper et al., 2012; Chadov et al., 2010; Hermet et al., 2013; Kieven et al., 2010; Fang et al., 2017). They possess high Currie temperature, semiconducting and semimetallic properties. The full Heuslers have a general formula X2YZ, while XYZ represents semi-Heusler systems. The Zatom belongs to the sp-block of the periodic-table, while Y and X are transitionmetals. The SHs possess a Cl<sub>b</sub> structure Corresponding author: <u>nenuwe.nelson@fupre.edu.ng</u>

with space group F-43m. The X, Y, Z atoms occupy the 4a (0, 0, 0), 4b (0.5, 0.5, 0.5), and 4c (0.25, 0.25, 0.25), with 4d (0.75, 0.75, 0.75) left vacant, while the FH occupies all four sites in the Wyckoff coordinates.

Recently, many SH alloys have been studied (Dey et al., 2021; Dutt et al., 2022; Enamullah et al., 2020; Osafile & Nenuwe, 2021; Parvin et al., 2021; Rogl et al., 2016; Sarwan et al., 2021). Semi-Heuslers are found to be thermally and mechanically stable and possess substantial thermoelectric power factor jaishi (Ahmad Khandy et al., 2021; Jaishi et al., 2021; Javed et al., 2020; Kieven et al., 2010; Rogl et al., 2016). These attributes qualify semi-Heusler materials for high-temperature

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thermoelectric (TE) applications.

Amazingly, many extrinsic semiconductors exhibit good TE behavior with values of figure of merit (zT) comparable to the wellknown silicon, lead, and bismuth-based TE compounds like SiGe, PbTe, BiSb BiSbTe, and Bi<sub>2</sub>Te<sub>3</sub> (Goldsmid & Douglas, 1954; Jaishi et al., 2021; Lee et al., 2012; Poudel et al., 2008; Poudeu et al., 2006). However, larger data of lattice thermal conductivity reduce the zT for intrinsic SH alloys. Studies have confirmed that the decline of thermal conductivity by doping of ZrCoBi (Zhu et al., 2018), TaFeSb (Zhu et al., 2019), and FeNbSb (Fu et al., 2015) SH alloys resulted in a significant improvement of the zT. Pressure application has also improved TE properties of SHs (Jaishi et al., 2021; Kumarasinghe & Neophytou, 2019).

Therefore, it is clear that SHs are good materials for TE applications, but some will require expensive dopants to diminish the lattice thermal conductivity and adjust the carrier concentration. Hence, further studies are needed to find efficient, nonharmful. economical, abundant and multifunctional materials (Jaishi et al., 2021). Also, SHs have attracted attention due to the possible adjustable band gap (Kumarasinghe & Neophytou, 2019), high optical absorption, and optical conductivity in the band range of the incident spectrum (Kieven et al., 2010; Wei & Wang, 2018). The noticeable band gap becomes useful in optoelectronic applications, and they act as a potential source to replace Pb-containing compounds (Eperon al., 2015; Ramasamy et al., 2016). From our literature review, we noticed very few theoretical studies on Co-based SH, and in particular, no published data was found for CoTiBi. Consequently, in this study, we utilised DFT to carry out a thorough examination of the structural, electronic, and phonon characteristics as well as the impact of mechanical. pressure on the thermodynamic, and elastic properties of CoTiSb and CoTiBi semi-Heuslers.

### 2. MATERIALS AND METHODS

The elastic, mechanical, thermodynamic, structural, and electronic properties of CoTiSb and CoTiBi semi-Heuslers were studied based on the density functional technique (Schwarz et al., 2002; Schwarz & Blaha, 2003; Sjöstedt et al., 2000). The full potential augmented plane waves plus local orbitals (FP-LAPW+lo) was used to approximate the Kohn-Sham equation, and generalized-gradient-approximation the was employed to solve the exchange correlation potential (Perdew et al., 1997). The elastic, mechanical, thermodynamic, and electronic properties were obtained from the WIEN2K computational package (Jamal et al., 2014), and the phonon dispersion properties were extracted from the Thermo-pw code implemented in Quantum Expresso (Malica & Dal Corso, 2021). In the WIEN2K, the input data such as *l*max, *k*-points, and *RK*max were chosen to be 10, 5,000, and 8.5, respectively. The k-points and Ecut in the Thermo pw package were optimized to (4 x 4 x 4) and 80, respectively.

### 3. RESULTS AND DISCUSSION

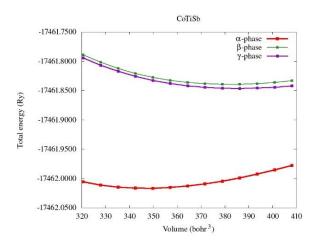
# 3.1 Structural and dynamical stability and electronic properties

The CoTiBi and CoTiSb semi-Heuslers crystalize in the cubic crystal structure in space group F-43m (No. 216) as a nonmagnetic semiconductor. From the wellknown Wyckoff atomic positions for the semi-Heusler alloys XYZ (4a(0, 0, 0), 4b(0.5, 0.5, 0.5), 4c(0.25, 0.25, 0.25)), three structural types are derived. We studied CoTiBi and CoTiSb alloys in three phases:  $\alpha = (4c, 4b, 4a), \beta = (4b, 4a, 4c),$ and  $\gamma = (4a, 4c, 4b)$ . The lattice constants and bulk modulus have been obtained by calculating the total energy for several volumes and fitted into the Murnaghan equation of states (Murnaghan, 1944; Tyuterev & Vast, 2006).

$$E - E_0 = \left[\frac{E_0}{B'} + \frac{V_0}{B'(B'-1)} + \frac{1}{(1-B')}\right]$$

B and B' represent bulk modulus and pressure derivative of B.  $E_0$  is the ground state energy, and  $V_0$  is the equilibrium volume. To determine the minimum energy for the  $\alpha$ -,  $\beta$ -, and  $\gamma$ -phases, as shown in Figure 1, we used Eq. (1) to plot the total energy against variation in volume. For both semi-Heuslers, the  $\alpha$ -phase was

discovered to have the lowest ground state energy, indicating that it is the most stable structural phase for both SHs. Table 1 reports the structural properties for the CoTiBi and CoTiSb SHs, and our findings are in conformity with the available data (Ma et al., 2017; Ouardi et al., 2012; Shi et al., 2017; Toboła et al., 1998).



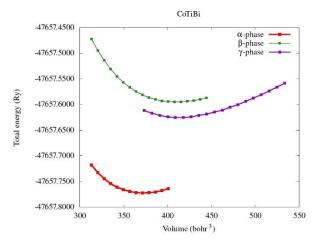


Figure 1: Calculated total energy versus volume for (Type 1, Type 2, Type 3) structural phases of CoTiSb and CoTiBi semi-Heuslers.

Table 1: Calculated data of lattice parameters  $a_0$  (Å), bulk modulus B (GPa), band gap  $E_g$  (eV), minimum energy  $E_m$  (Ry), and formation energy ( $E_f$ ).

Semi-	Phase	<b>a</b> 0	$E_{ m m}$	$oldsymbol{E}_{\mathbf{f}}$	В	$oldsymbol{E}_{\mathbf{g}}$
Heuslers						
	α	6.04	-47,657.772783	-0.048	122.06	0.91
CoTiBi	β	6.25	-47,657.625283	0.0012	90.00	-
	γ	6.24	-47,657.595424	0.0111	95.76	-
CoTiSb	α	5.90	-17,462.016904	-0.069	142.63	1.05
		5.88		-0.67	142	1.06
		5.88				0.95
		5.90				1.03
	β	6.09	-17,461.839494	-0.01	110.42	-
	γ	6.11	-17,461.846274	-0.012	102.75	-

References: (Ouardi et al., 2012; Ma et al., 2017; Tobola et al., 1998; Shi et al., 2017).

Also, the formation energy was computed using the relation in Eq.(2) (Z. W. Huang et al., 2012; Yakoubi et al., 2012) to establish

the structural stability and feasibility of the studied SH compounds.

$$E_f = \frac{E_T - xE_{C_0} - yE_{T_i} - zE_{B_i(Sb)}}{(x + y + z)}$$

where  $E_{\text{Co}}$ ,  $E_{\text{Ti}}$ ,  $E_{\text{Bi}}$ , and  $E_{\text{Sb}}$  are the total energies of the pure elements Co, Ti, Bi, and Sb, respectively;  $E_{\text{T}}$  is the total energy of the SH alloy; and x, y, and z are the numbers of Co, Ti, Bi, and Sb atoms in the primitive cell, respectively. Table 1 displays the values of minimal energy, energy of individual elements, and resultant formation energy. We discovered that the  $\alpha$ -phase formation energy is negative, indicating that the formation and feasibilty of CoTiBi and CoTiSb are possible.

Figure 2 depicts the phonon dispersion and phonon density of states for CoTiBi and

(2)

CoTiSb. There are six optical and three acoustic modes in these semi-Heuslers since each unit cell has three atoms. The fact that all vibration modes in Figure 2 are positive confirmed the dynamical stability of these compounds. For CoTiSb and CoTiBi, respectively, the highest optical branches have frequencies of 312.51 cm<sup>-1</sup> and 264.04 cm<sup>-1</sup>. We noticed that, due to an increase in atomic mass, the frequency of CoTiSb to CoTiBi decreased (Guo et al., 2015). The non-appearance of imaginary frequencies in the phonon scattering curve shows that CoTiBi and CoTiSb are thermodynamically stable materials.

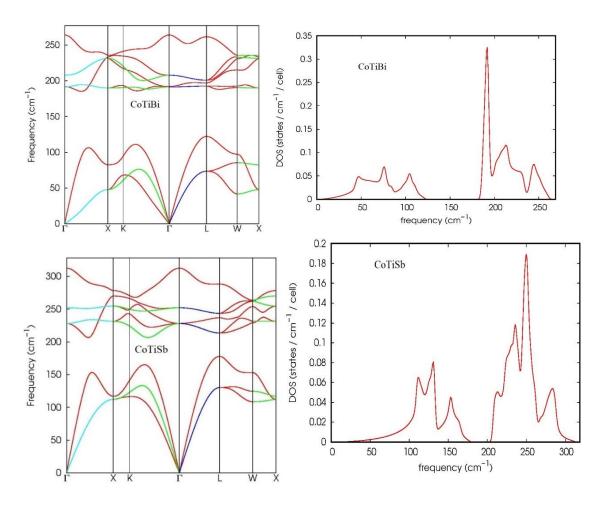


Figure 2: Phonon dispersion and phonon density of states for CoTiBi and CoTiSb.

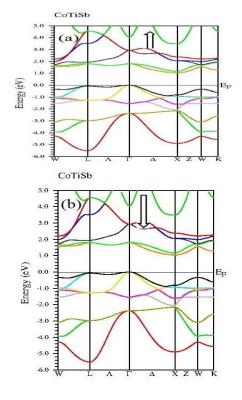
The considered electronic bandstructures

along the high symmetry k-points are

presented in Figure 3. According to (Kaur & Kumar, 2017), the presence of an electronic band in semi-Heuslers with 18valence electrons is mainly due to *d*-orbitals. In Figure 3, the conduction band (CB) minimum is located at X high symmetry point, and the valence band (VB) maximum is located at  $\Gamma$  point, which implies that both CoTiBi and CoTiSb are indirect band gap semiconductors with an energy gap of 0.91 and 1.05 eV, respectively. The calculated band gap for CoTiSb is (1%) less than that reported in Ref. (Ouardi et al., 2012), in line with the fact that the band gap obtained from GGA underestimates experimental values. Also, our result is 9.5% and 1.9% higher than the results obtained using the Korringa-Kohn-Rostoker method (Toboła et al., 1998) and Heyd-Scuseria-Ernzerhof (HSE06) hybrid functional (Shi et al., 2017), as shown in Table 1. However, the literature has no data on band gap and lattice parameters to compare results obtained for CoTiBi alloy. Therefore, results obtained for CoTiBi semi-Heusler can be used as a

reference for further studies.

The projected density of states (PDOS) and total density of states (TDOS) for CoTiSb and CoTiBi, respectively, are calculated and shown in Figures 4 and 5, respectively, to analyse the contributions of various atomic orbitals to the band gap. Figures (4a) and 5a) show that for both semi-Heuslers, the d, d-eg, d-t2g orbitals of the Y and Xatoms (Ti and Co) make majority contributions to the energy gap from both VB and CB. The TDOS: Figures (4b and 5b) show that both the up-spin and down-spin channels in the spin-polarized state of these materials have an energy gap, proving that CoTiSb CoTiBi and are pure semiconductors channels. in both Additionally, we noticed that the Fermi level ( $E_f$ ), which is known to move towards the CB for n-type and VB for p-type extrinsic semiconductors, shifts towards the valence bands in Figures (3, 4 & 5), indicating that CoTiSb and CoTiBi semi-Heuslers are likely to function better as ptype semiconductors (Jhi et al., 1999).



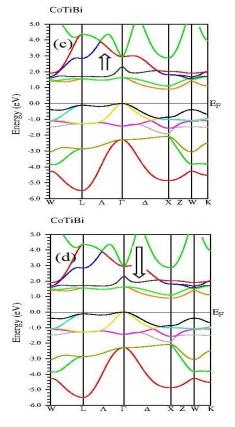


Figure 3: Bandstructure of up spin (a, c) and down spin (b, d) for CoTiSb and CoTiBi

### semi-Heuslers.

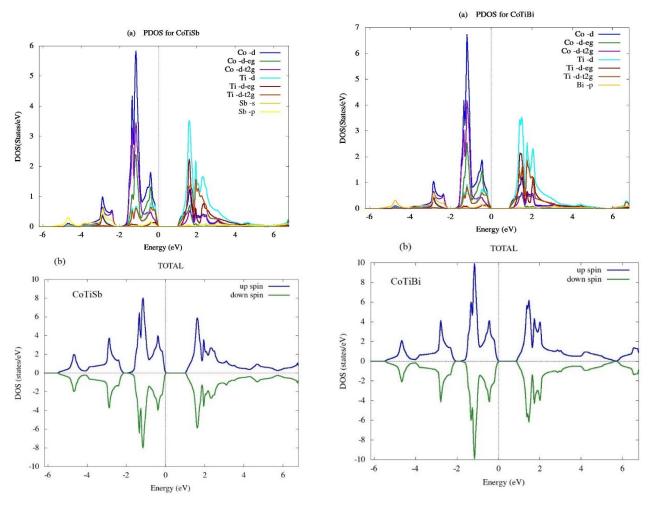


Figure 4: (a) Projected density of states, and (b) Total density of states for CoTiSb semi-Heusler.

Figure 5: (a) Projected density of states, and (b) Total density of states for CoTiBi semi-Heusler.

### 3.2 Elastic constants

Elastic constants are crucial properties of solid state materials. They are employed to ascertain how a crystal reacts to external forces and provides crucial information about the types of forces operating in a crystal structure (Mehl, 1993). Also, elastic constants can provide important data such as Young, bulk, and shear moduli, Poisson ratio, Pugh ratio, isotropic factor, stiffness, ductility, stability, and brittleness of materials. In order to stay within the elastic domain of the crystal structure, only small lattice distortions were used, leading to the determination of the second-order elastic constants ( $C_{ij}$ ) for cubic CoTiBi and

CoTiSb compounds at 0 K and zero pressure (Wang et al., 1993). A cubic crystal structure is known to have three different elastic constants ( $C_{11}$ ,  $C_{12}$ ,  $C_{44}$ ). Table 2 contains the elastic constants of CoTiBi and CoTiSb that were calcuated for this study. According to the Born stability conditions, the cubic elastic constants must satisfy the following requirements (Wang et al., 1993) in Eq. (3). All calculated elastic constants of CoTiBi and CoTiSb semi-Heuslers satisfy Eq. (3), indicating that these semi-Heusler alloys are mechanically stable. The results in Table 2 agree with the values reported by Siham Ouardi et al. (Ouardi et al., 2012),

where they studied the transport and mechanical behaviors of pure and doped CoTiSb compounds.

 $C_{11} - C_{12} > 0$ ,  $C_{44} > 0$ , and  $C_{11} + 2C_{12} > 0$  (3)

Table 2: Calculated values of the elastic constants  $(C_{ij})$ , Young's modulus (E), bulk modulus (B), and shear modulus (G) in GPa, Poisson ratio (v), isotropic factor (A), Pugh ratio, Cauchy

pressure and melting temperature  $(T_{\rm m})$  for CoTiBi and CoTiSb semi-Heuslers.

Compou	C <sub>11</sub>	C <sub>12</sub>	C44	В	$\boldsymbol{G}$	E	V	$\boldsymbol{A}$	B/G	$C_{12}$ -	$T_{\rm m}\left({ m K}\right)$
nd										C44	$\pm 300$
CoTiBi	221.9	73.3	75.1	122.8	74.7	186.5	0.25	1.01	1.6	-1.77	1864.6
	3	4	1	8	8	1			4		3
CoTiSb	258.5	75.2	95.8	136.3	94.1	229.6	0.22	1.05	1.4	-20.66	2028.7
	0	2	8	1	6	1	0.24	1.06	5	-9	4
	259	84	93	142	91	224			1.5		-
									6		

Reference: (Ouardi et al., 2012).

It is well established that pressure has a considerable impact on the elastic and mechanical properties of materials (Hill, 1952; Jiao et al., 2013; Korozlu et al., 2013; Nelson and Judith, 2021; Nenuwe and Agbalagba, 2021; Sailuam et al., 2022) (Wani et al., 2021). In this study, we investigated how pressure affects mechanical and elastic characteristics of CoTiBi and CoTiSb SHs. At applied hydrostatic pressure in the range of (5 - 30) GPa, we presented the elastic constants of CoTiBi and CoTiSb semi-Heuslers in Table 3 in steps of 5 GPa. To the best of our knowledge, the elastic constants of CoTiBi and CoTiSb under pressure have not been published in the literature. Also, we plot the pressure-dependent behavior of elastic constants of CoTiBi and CoTiSb SH as shown in Figure 6. We observed that pressure significantly enhanced the elastic constants  $C_{ij}$ , and the values of  $C_{11}$  are higher than  $C_{12}$ , and  $C_{44}$  within the entire range of pressure considered. The  $C_{12}$ , and  $C_{44}$  are shear elastic constants and signify elasticity in shape. While  $C_{11}$  signifies elasticity in length. The calculated  $C_{11}$  and

 $C_{44}$  for CoTiSb are higher than those of CoTiBi, which suggests that the strength of CoTiSb is higher than CoTiBi alloy. The Cauchy pressure  $(C_{12} - C_{44})$  provides information about bonding in a cubic material. A negative Cauchy pressure ( $C_{44}$ >  $C_{12}$ ) signifies a covalently bonded solid, while positive Cauchy pressure ( $C_{44} < C_{12}$ ) represents ionic bonding. Also, Cauchy pressure predicts that positive values indicate ductility while negative values indicate brittleness for a given material. The data at zero pressure in Table 2 shows that these compounds are covalently bonded and are brittle. This agree with the negative value reported in REF. (Ouardi et al., 2012) for CoTiSb. In Table 3, the Cauchy pressure is negative as pressure rises from 5 -15 GPa and 25 - 30 GPa. At the pressure value of 20 GPa, the Cauchy pressure was positive, implying that CoTiBi compound is brittle when pressure rises from 0 - 15 GPa, and 25 – 30 GPa. Moreover, at 20 GPa, CoTiBi behaves like a ductile material, which may be associated with a phase transition that might have occurred around 20 GPa pressure

Materials	Pressure	C <sub>11</sub>	$C_{12}$	C44	$C_{12}$ - $C_{44}$
	5	258.81	88.00	88.55	-0.55
	10	298.90	98.66	105.07	-6.41
CoTiBi	15	332.86	110.92	118.95	-8.03
	20	372.79	133.75	130.96	2.79
	25	408.62	144.14	145.33	-1.19
	30	437.85	151.78	159.68	-7.9
	5	297.51	96.84	107.17	-10.33
	10	332.98	105.04	116.88	-11.84
CoTiSb	15	366.04	115.60	127.01	-11.41
	20	396.29	124.32	139.82	-15.5
	25	431.0	134.39	154.78	-20.39
	30	456.88	141.19	166.75	-25.56

Table 3: Calculated values of the elastic constants  $C_{ij}$  (GPa) and Cauchy pressure under different pressure for CoTiBi and CoTiSb semi-Heuslers

### 3.3 Mechanical properties

The bulk modulus (B), Young modulus (E), shear modulus (G), Poisson ratio (v), isotropy factor (A), and Pugh ratio are important mechanical behaviors of

materials that can be calculated from the elastic constants of solid materials using the Voigt-Reuss-Hill average technique (Huang et al., 2014).

$$B = \frac{1}{3}(C_{11} + 2C_{12})$$

$$B_{V} = B_{R} = B$$

$$G = \frac{1}{2}(G_{v} + G_{R})$$

$$G_{R} = \frac{5(C_{11} - C_{12})C_{44}}{3(C_{11} - C_{12}) + 4C_{44}}$$

$$(7)$$

$$G_{v} = \frac{1}{5}(C_{11} - C_{12} + 3C_{44})$$

$$E = \frac{9BG}{3B + G}$$

$$(9)$$

$$A = \frac{2C_{44}}{C_{11} - C_{12}}$$

$$(10)$$

$$v = \frac{3B - 2G}{2(3B + G)}$$

The Reuss and Voigt approximations of G and B are respectively,  $G_R$ ,  $G_v$ ,  $B_R$ , and  $B_v$ . Table 4 provides a list of the computed mechanical properties. Shear modulus (G) demonstrates how resistant a material is to plastic deformations under shear stress, while the bulk modulus (B) measures a material's resistance to volume changes brought on by applied pressure (Ali &

Rahaman, 2018; Boucetta, 2014). Figure 7 shows how pressure affects *E*, *G*, and *B*. As can be observed, all the parameters respond virtually linearly to changes in external pressure. According to Figure 7, the values of the bulk modulus *B* and shear modulus *G* for CoTiBi and CoTiSb, respectively, at a pressure of 5 GPa are 144.94 and 87.27 GPa and 163.73 and 104.38 GPa, respectively. *B* 

and *G* have 247.14 and 152.79 GPa for CoTiBi and 246.42 and 163.13 GPa for CoTiSb when the pressure hits 30 GPa. The findings of the current calculations show that the bulk modulus is more pressure sensitive than the shear modulus. A significant quantity is the Young's modulus *E*, which is the proportion of tensile stress

to tensile strain (Pugh, 1954). It serves as a gauge for determining solid rigidity. The higher the value of E, the more rigid the material is. Table 4 shows that the computed value of E grows practically linearly with pressure, demonstrating that the pressure significantly affects the stiffness of CoTiBi and CoTiSb compounds.

Table 4: Calculated values of Young's modulus (E), bulk modulus (B), v and shear modulus (G) in GPa, Poisson ratio (v), isotropic factor (A), and Pugh ratio under different pressure for CoTiBi and CoTiSb semi-Heuslers

Materials	Pressure	В	G	E	B/G	V	A
	5	144.94	87.27	218.06	1.66	0.25	1.03
	10	165.41	103.06	256.01	1.60	0.24	1.05
CoTiBi	15	184.90	115.69	287.17	1.60	0.24	1.07
	20	213.43	126.25	316.38	1.69	0.25	1.10
	25	232.30	140.94	349.62	1.64	0.25	1.10
	30	247.14	152.79	380.07	1.62	0.24	1.12
	5	163.73	104.38	258.26	1.57	0.24	1.07
	10	181.02	115.71	286.15	1.56	0.24	1.03
CoTiSb	15	199.08	126.29	312.74	1.58	0.24	1.01
	20	214.97	138.28	341.59	1.56	0.24	1.03
	25	233.26	152.15	374.93	1.47	0.23	1.04
	30	246.42	163.13	400.92	1.51	0.23	1.06

The bulk modulus to shear modulus (B/G)ratio defines a material plastic characteristic. According to the Pugh criterion (Greaves et al., 2019), the difference between ductile and brittle material is 1.75, which is the critical value. A high ratio of B/G indicates ductility, whereas a low number indicates the material's brittleness. In Table 4, the calculated results are displayed. As can be observed, all values of B/G are below the critical value of 1.75, demonstrating the brittle behavior of CoTiBi and CoTiSb alloys. The impact of pressure on the value of B/G is depicted in Figure 8. We observed that B/G decreased as pressure rises, suggesting that pressure can reduce the brittleness of these compounds. In addition, Poisson's ratio is also used to classify materials' bonding nature. The range of values lies between 0 < v < 0.5. Values higher than 0.25 corresponds to materials associated with central forces and show

better plasticity, whereas values lower than 0.25 correspond to materials dominated by covalent bonding (Zener & Siegel, 1949). Our current work shows that around the pressure values of 0, 5, 20, and 25 GPa, central forces are operational in CoTiBi, and at 10, 15, and 30 GPa, its dominated by covalent bonding. For CoTiSb, the values of v from 0 - 30 GPa indicate covalent bonding in this material. The degree of anisotropy in the solid structure is mostly determined by the isotropy factor (Khandy & Gupta, 2016). When A = 1, the compound is entirely isotropic. As shown in Table 4, both compounds have anisotropy factors that are A > 1, showing that they are elastically anisotropic. In Figure 8, the impact of pressure on the values of A is also reported. The value of A increases monotonically with pressure for CoTiBi, while, for CoTiSb it initially decreased and increased with a rise in pressure.

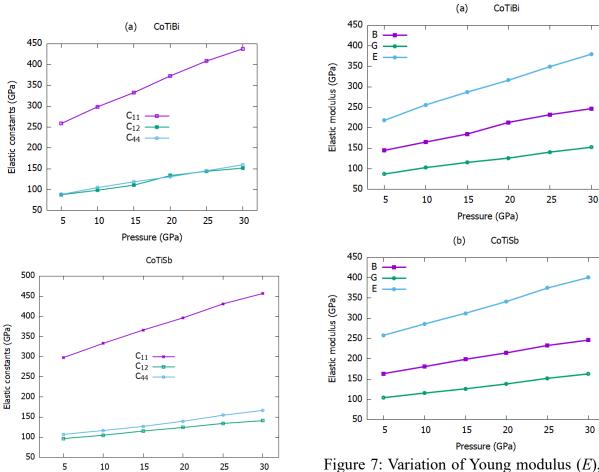
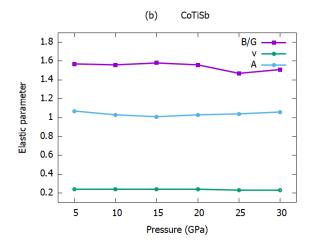


Figure 6: Variation of elastic constants ( $C_{11}$ ,  $C_{12}$ ,  $C_{44}$ ) with pressure for (a) CoTiBi and (b) CoTiSb.

Pressure (GPa)

Figure 7: Variation of Young modulus (E), shear modulus (G), and bulk modulus (B) with pressure for (a) CoTiBi and (b) CoTiSb.



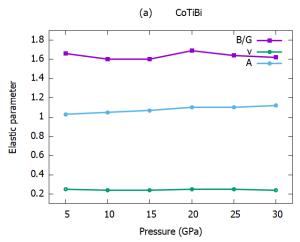


Figure 8: Variation of Pugh ratio, Poisson ratio (v), and isotropy factor (A) with pressure for (a) CoTiBi and (b) CoTiSb.

### 3.4 Thermodynamic properties

From the calculated elastic constants, we have determined the Debye temperature  $(\Theta_D)$  and other related parameters at

different pressures using the equations below (Khandy et al., 2018; Khandy & Gupta, 2016).

$$\Theta_D = \frac{h}{k_B} \left(\frac{3N}{4\pi V}\right)^{\frac{1}{3}} v_a \tag{12}$$

Where  $k_B$ , V, h, and N are the Boltzmann constant, volume, Planck's constant, number of atoms per formula unit. The mean sound velocity  $v_a$  calculated from the equations:

$$v_a = \sqrt[3]{3} \left( \frac{2}{v_t^3} + \frac{1}{v_l^3} \right)^{-\frac{1}{3}} \tag{13}$$

The transverse ( $v_t$ ) and longitudinal ( $v_l$ ) elastic sound velocites of CoToSb and CoTiBi are calculated from the bulk and shear modulus in Navier's equation (Schreiber et al., 1975).

$$v_l = \left(\frac{3B + 4G}{3\rho}\right)^{\frac{1}{2}} \tag{14}$$

$$v_t = \left(\frac{G}{\rho}\right)^{\frac{1}{2}} \tag{15}$$

The melting temperature and Gruneisen parameter calculated from Poisson ratio v are given as  $T_m = (553 \text{ (K)} + (5.911) \text{ C}_{11}) \pm 300 \text{ K}$  (16)

$$\gamma = \frac{3(1+v)}{2(2-2v)} \tag{17}$$

The Debye temperature is a very significant quantity in material science, which plays a crucial role in several properties of solid materials, such as thermal vibration of atoms, specific heat, melting point, and many other properties of solids (Boucetta, 2014; Qi et al., 2015). It also predicts the strength of bonds and the structural stability in solids. In order to gain a deeper insight

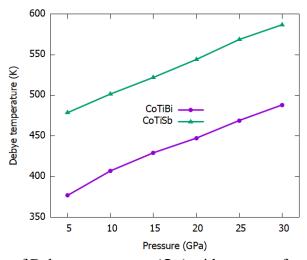
into the lattice stability and thermodynamic properties under pressure, it is necessary to compute the values of  $\Theta_D$  under different pressure. Employing Eqs. (13) – (15), the results obtained are reported in Table 5. The pressure dependence of  $\Theta_D$  for both compounds is displayed in Figure 9. Values obtained reveal that the Debye temperature  $(\Theta_D)$  increases with rising pressure for both

CoTiBi and CoTiSb. The values of  $\Theta_D$  at 5 and 30 GPa for CoTiSb are 377.02 and 488.27 K, respectively, and for CoTiSb are respectively 478.71 and 586.79 K. We observed that CoTiSb is more sensitive to pressure and has the highest  $\Theta_D$ . Therefore,

with rising pressure, the covalent bonds get stronger. CoTiSb possesses a stronger covalent bond and high thermal conductivity than CoTiBi semi-Heusler alloy. The melting temperature  $(T_{\rm m})$  was observed to increase monotonically with rising pressure for both CoTiSb and CoTiBi.

Table 5: Calculated pressure dependence of transverse sound velocity  $v_t$  (m/s), longitudinal sound velocity  $v_t$  (m/s), average sound velocity  $v_a$  (m/s) calculated from elastic moduli, Debye temperature  $\Theta_D$  (K), and Gruneisen parameter ( $\gamma$ ) under different pressure for CoTiBi and CoTiSb semi-Heuslers

Materials	Pressure	$v_t$ (m/s)	$v_l$ (m/s)	$v_a$ (m/s)	$\Theta_D(K)$	Γ	$T_{\rm m}({ m K}) \pm 300$
	5	2973.11	5144.44	3300.42	377.02	1.5	2,082.82
	10	3180.05	5451.05	3527.19	407.21	1.45	2,319.79
CoTiBi	15	3323.24	5690.04	3685.64	429.41	1.45	2,520.53
	20	3429.84	5964.13	3809.09	447.40	1.5	2,756.56
	25	3571.94	6179.87	3965.14	469.11	1.5	2,968.35
	30	3695.85	6348.65	4100.08	488.27	1.45	3,141.13
	5	3698.67	6300.71	4100.11	478.71	1.45	2,335.38
	10	3840.11	6537.01	4256.62	501.63	1.45	2,521.24
CoTiSb	15	3962.93	6759.91	4393.58	522.03	1.45	2,716.66
	20	4101.48	6970.08	4545.62	544.07	1.45	2,895.47
	25	4259.89	7212	4719.53	568.63	1.41	3,100.64
	30	4371.19	7371.53	4841.06	586.79	1.41	4,715.17



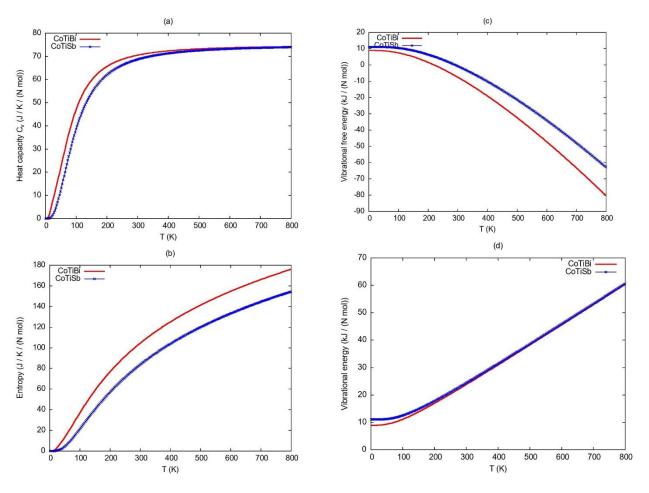
**Figure 9:** Variation of Debye temperature  $(\mathbf{O}_D)$  with pressure for CoTiBi and CoTiSb.

Thermodynamic variables including Debye free energy (F), Debye entropy (S), Debye vibrational energy (E), and Debye heat capacity  $(C_v)$  have also been determined.

Figure 10(a-d) displays the estimated  $C_v$ , S, F, and E in temperature range from 0 to 800 K. The specific heat of a solid can be used to describe how different thermodynamic

restrictions affect the behaviour of various materials. It also specifies how well the substance can hold heat. As a function of temperature, the phonon contribution dominates  $C_v$ . Figure 10(a) demonstrates that the Debye heat capacities  $C_v$  of CoTiSb and CoTiBi, obeys the Debye power law model,  $C_v \propto T^3$  (Debye & Ann, 1912), as anticipated. This model accurately predicts heat capacity at low temperatures. At room temperature (298 K), it was found that the heat capacities of CoTiSb and CoTiBi respectively grew quickly to around 68.6 J/K/mol and 70.43 J/K/mol. And as the temperature increases,  $C_{\rm v}$  converges to a constant value of 73.94 J/K/mol for CoTiSb

and 74.22 J/K/mol for CoTiBi, indicating a strong agreement with the Dulong-Petit law Dulong, 1981) (Petit and at temperatures. Since thermal agitation causes disorder, the entropy grows as the temperature rises. Figure 10(b) illustrates the entropy behavior with temperature. Figure 10(c) depicts the Debye free energy (F) of CoTiSb and CoTiBi, which decreases as temperature rises. The falling trend of F is extremely typical and gets worse as any natural process progresses. The entropy of any system determines the degree of F. Figure 10(d) illustrates a rising trend in the Debye vibrational energy (E) temperature.



**Figure 10:** (a) Heat capacity  $(C_v)$ , (b) entropy, (c) vibrational free energy, and (d) vibrational energy in the temperature range of (0 - 800) K.

### 4. CONCLUSION

The first principle computation using

WIEN2K was succefully emploed to optimize the structural properties, and study the phonon, electronic, thermodynamic,

and mechanical properties of CoTiSb and CoTiSb semi-Heusler alloys. The lattice constants of the semi-Heuslers were found to be 5.9 and 6.04 for CoTiSb and CoTiBi, respectively. Both alloys were found to be mechanically stable at zero and applied hydraustatic pressure by satisfying the wellestablished Born stability conditions. The Pugh ration of CoTiSb and CoTiBi was found to be less than 1.75, which indicates that these compounds are brittle in nature even at zero and applied hydraustatic elastic constants pressure. The mechanical parameters were observed to be significantly influenced by presssure. The results obtained from the phonon dispersion curves confirmed that the alloys are

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thermodynamically stable. The heat capacity  $C_v$  was found to converge at 73.94 JK<sup>-1</sup>mol<sup>-1</sup> and 74.22 JK<sup>-1</sup>mol<sup>-1</sup> for CoTiSb and CoTiBi, reflecting the well-esblished Dulong-Petit law. From the elctronic properties, CoTiSb and CoTiBi were found to be an indirect band gap semiconductor with energy gap of 1.05 eV and 0.91 eV. The results obtained for elastic constants, lattice parameters, and energy gap for CoTiSb are in agreement with available data. However, there are no data to compare the results predicted for CoTiBi semi-Heusler. Therefore, the predicted results for CoTiBi may serve as reference to further studies.

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