Research Article / Review Article



FUPRE Journal of



Scientific and Industrial Research

ISSN: 2579-1184(Print)

http://fupre.edu.ng/journal

ISSN: 2578-1129 (Online)

Synthesis and characterization of copper doped TiO₂ via electrodeposition method for

photovoltaic applications

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ARTICLE INFO

Received: 10/07/2024 Accepted: 21/10/2024

Keywords

Electrodeposition, Doping, Nanoparticles, Titanium dioxide

ABSTRACT

Indium doped Tin Oxide (ITO) substrate was used to synthesize Cu and Cu-TiO₂ nanocrystals via electrochemical deposition method. The techniques used for analyzing the structure, morphology, optical, magnetic and electrical properties include X-ray diffractometry, scanning electron microscopy, energy dispersive X-ray (EDAX), UV-visible spectrophotometer, It was discovered that the energy bandgap shrank from 3.3 - 3.1 eV. The magnetic analysis showed a characteristic ferromagnetic ordering, with 5.13×10^{-4} emu/g for the 3% and 6.08×10^{-4} emu/g for the 5% samples, respectively. The maximum conductivity of $0.08 \ \Omega \text{cm}^{-1}$ was recorded by the 5% doped thin film.

1. INTRODUCTION

Titanium dioxide (TiO₂) is a transparent conducting oxide which belongs to group II-VI semiconductor material. TiO₂ has a wide bandgap ($E_g \ge 3 \text{ eV}$) and high refractive index which makes it potent in the field of science and industries Zeribi et al., (2022), Kim et al., (2015) and Tipparak, et al., (2017). Naturally TiO₂ has high resistivity value near $10^8 \Omega$ cm and with excess titanium ions ultimately alters the stoichiometric composition which results to n-type semiconductor. According to Mariatz et al., (2017) the distortion of the arising from crystal the change in composition of oxygen to titanium ratio

culminates into enhancement of its electrical properties TiO₂ thin film has attracted attention of researchers for some decades due to the physical, electrical and chemical properties which has enhanced its applications for photonic device and solar cell, photo-catalysis, gas sensing, optical filters and anti-reflection coatings (Haque et al., 2019). The photoactivity property of TiO_2 is well exhibited by the anatase and rutile polymorphs under solar light irradiation according to Comert et al., (2016). It has also been recorded amongst researchers that TiO₂ has high recombination rate of electron hole pairs and absolves ultra violet light in low

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wavelength region. However, the utmost performance of TiO₂ can be obtained by structural modification in terms of doping with transition metals (Tipparak et al., 2017). It is obvious that copper has a high absorption coefficient with relatively narrow band gap which easily reduces the bandgap of the host atom (karabulut et al., 2018). Copper dopant titanium lattice enhances electrical in conductivity reducing by electron recombination and having a similar ionic radius to titanium (Thangama et al., 2017, Khan et al., 2017). It is certain that dopants such as cobalt, chromium, nickel, iron, etc have been used to dope titanium oxide adopting several deposition techniques. The work centered on doping of TiO₂ with copper using electrodeposition method in order to study the optical, electrical and magnetic properties as work done by other researchers were limited to the optical properties of TiO₂ using other methods of deposition. The electrodeposition method was preferred amongst other methods due to its cost effectiveness. simplicity, environmental friendliness (Bensouici et al., 2016). Many deposition techniques that have been employed for the doping of copper on titanium thin films include sol-gel (Dharmadasaz and Haigh, 2006), Spray pyrolysis (Khan et al., 2017), Hydrothermal (Patel and Gajbhiye, 2012), Chemical Vapour Deposition (Fasakin et al., 2013). Furthermore, Perarasan et al., (2019) adopted chemical hydrolysis technique to synthesize Cu doped titanium dioxide nanoparticles. The XRD technique revealed tetragonal structure with anatase phase and the SEM results depicted nanoparticles of spherical morphology. The optical absorbance showed increase of bandgap with percentage doping. The optical properties of Cu-TiO₂ was studied by Álvaro et al., (2017) using green synthesis technique. The spherical shape of the particles and their average size of 27.3 to 70 nm were revealed through SEM imaging.

As the amount of Cu increased, the band gap energy showed a reduction. Ahmed et al., (2017) used the inert gas condensation technique to analyze the structural and optical properties of Cu-TiO2. The XRD patterns clearly indicated the presence of rutile phase peaks in both the undoped and Cu-doped TiO₂ thin films. As copper concentration increased, the energy band gaps showed a decrease. Also Khan et al., (2017) used sol-gel spin coating method to investigate the electrical properties of Cu doped TiO_2 . The morphology of the thin films showed that the number of layers increased proportionately with the grain size and the resistivity decreased with increasing layer. TiO₂ is a valuable material that absorbs photons and readily converts them into electric current and high value of transmittance of titanium dioxide makes it suitable for photovoltaic applications as window layer.

2. Experimental Procedure

2.1 Materials

The substrate used is indium-doped tin hydroxide, oxide.. sodium titanium trichloride (TiCl₃) citric acid as buffer. sulphate pentahydrate copper solution (CuSO₄.5H₂O) and, in the process of electrochemical deposition for undoped and films, silver monochloride Cu-TiO₂ electrodes were used.

2.2. Synthesis

Electrodeposition was utilized to produce $Cu-TiO_2$ and undoped TiO_2 thin films. To ensure strong adhesion to the ITO and promote material growth, the ITO coated substrates were cleaned with detergent, rinsed, and exposed to acetone for 30 minutes. Then 100 ml of 0.5 M titanium trichloride (TiCl₃) was mixed with 3% and 5% weight of CuSO₄.5H₂O solution at room temperature to prepare the solutions of the undoped TiO₂ and 60 ml of 1 mol of NaOH was mixed with 100 ml of TiCl₃. For each of

(1)

the depositions, an ITO-coated substrate was positioned upright in a three-electrode chamber that contained the anode, working electrode, and reference electrode. On the other hand, the deposition was done for 5 minutes at a potentiostatic state of -200 mV. Following the completion of deposition, both undoped and Cu-TiO₂ thin films were subjected to cleaning, drying, and annealing at 300°C for 30 minutes.

2.3 Characterization

Figure 1 illustrates the diffraction patterns of both undoped TiO₂ and Cu-TiO₂ films. Philips diffractometer model PW1800 was used to characterized the XRD patterns of the annealed films, revealing crystallographic phases between $10^{\circ}C \le 2\theta \le 70^{\circ}C$. The films' surface micrograph was examined using a JEOL JSM 35 CF SEM, and the copper doped films' elemental composition was analyzed with EDAX. The JASCO 670, a double beam UV-Vis, NIR spectrophotometer, was utilized to analyze optical properties between 300 and 1000 nm in wavelength. Using a VSM (Lake Shore Model 7404), the magnetization data was collected, while the resistivity of the thin films was measured using a Keithlev 182 Nanovoltmeter.

3. Result and discussions

3.1. Phase studies

Fig. 1 displays the diffraction patterns of undoped and copper-doped TiO₂ films. The observed pattern of several peaks diffracted at 29.35°, 33.94°, 35.11°, 50.12° and 65.58° which confirmed anatase phase assigned to (101), (110), (220), (022), (123) crystal planes respectively. The peaks are of polycrystalline phase with high intensity peaks for the doped samples at (101), 110, (220), (022), (123) orientation which agrees with JCPDS file. There is little difference in diffraction patterns between the undoped and doped samples. This might be due to the small concentration of Cu⁺² that dissolved into TiO₂ crystal lattice having almost same ionic radii. The works of Perarasan et al., (2019) and Sokoidanto et al., (2020) also confirmed anatase phase resulting to increase in crystallite sizes of TiO₂ when doped with copper. The average crystallite sizes of the undoped and doped films were calculated using Scherer equation as shown in equation.

$$D = \frac{\kappa\lambda}{\beta Cos\theta}$$

The variables in equation 1 are defined as follows: λ is X-ray wavelength, d is crystallite size, θ is diffraction angle, and β is full width of half maximum.

3.2 Surface micrograph of the material

The surface morphologies of undoped and copper doped titanium dioxide films generated at room temperature is as illustrated in Figure 2.1(a-c). There is agglomeration of cross-linked, grey and irregular spherical nanostructures in the 5% doped sample than the 3% doped and the undoped. The undoped sample is denser than the 3% and 5% doped samples as illustrated in Figure 2. (a). The effect of doping is more pronounced in the 5% doped sample than the 3% doped and undoped thin films. This indicates the presence of cupper in the lattice crystals of titanium dioxide. This same result that showed increase in doping percentage without cracks was also obtained by Zhang et al., (2004) and Natsir et al., (2021). The EDAX images depicted in Fig. 2 (a-c) depicts the characteristic peaks of Ti, O and Cu elements. The EDX analysis performed was to find out the percentage of the elemental constituents arising from the synthesis. The characterization result showed the peak of Ti at 2.0 KeV, the peak of O at 0.31 KeV and the peak of Cu at 2.5 KeV as shown in Fig. 2. (a–c).



Fig. 1. XRD of TiO_2 for the undoped and doped samples



Fig. 2. SEM images of (a) undoped TiO₂, (b) 3% Cu:TiO₂ and (c) 5% Cu:TiO₂



Fig. 3. EDAX spectra of (a) undoped TiO_2 , (b) 3% Cu: TiO_2 and (c) 5% Cu: TiO_2

3.3 Optical studies

Figure 4 (a & b) displays absorbance and transmittance spectra in the 300-1000 nm optical range at room temperature. By employing a UV-visible spectrophotometer, we were able to quantify the absorbance and transmittance. Both undoped and doped films show a decrease in absorbance as the wavelength increases in the visible region. At a wavelength of 450 nm, the 5% doped thin film exhibited an absorption of 33.6%, while the 3% and undoped thin films had absorptions of 31.5% and 28.2% respectively, as depicted in Fig. 3a. This result is in accordance with the works of Ahmed et. al., (2017). It is observable that within the wavelength range of 450nm to 800nm, the transmittance spectra showed that the undoped and 3% doped thin films transmitted highest values of 56% to 83%. This result is attuned with the works of Khan et al., (2017).

3.4 Optical band gap

The plot of absorption coefficient versus photons energy as shown in Fig.5 is used to

determine the optical band gaps of the undoped and Cu-doped thin films. The optical bandgaps were calculated using Taucs relation as shown in equation (2).

 $(\alpha h\nu)^2 = A(h\nu - E_g)^n$ (2)

The absorption coefficient A depicts the constant that relies on the photon energy hv and the band gap energy Eg. As depicted in Fig. 5, the energy band gap decreases with an increase in doping percentage. Copper has the ability to modify the absorption properties of titanium by absorbing visible light and introducing electron states to the lattice structure. A change in local levels density causes a decrease in optical energy gap values. The decrease in energy bandgap values may be a result of the increase in grain size. In the undoped TiO₂ thin film, the direct band gaps were determined to be 3.32 eV, whereas the 3% and 5% doped films displayed bandgaps of 3.18 and 3.10 eV, respectively. This result is in concordant with the works of Suber et al., (2021) and Bedikvan et al., (2013). Also Heiba et al.,

(2022) worked on copper doped titanium dioxide and got the bandgap energy values that ranged from 3.18 to 3.38 eV for higher

doping percentage of 5-10% while Natsir et al., (2021) calculated the band gap of copper doped titanium dioxide as 3.10 eV.



Fig. 4. [a] Absorbance and [b] Transmittance spectra for the undoped and doped TiO₂: Cu films



Fig. 5. Plot of bandgaps of the materials.

3.5. Refractive index

Refractive index

The refractive indices for the doped and $Cu:TiO_2$ were obtained using equation (3) [31]:

(3)
$$\eta = (1 + \sqrt{R})/(1 - \sqrt{R})$$

The refractive index is represented by η , and reflectance is denoted by R. Figure 6 illustrates the relationship between photon energy and refractive index for undoped and Cu-TiO₂ thin films. Fig 6 shows that the refractive index values rose at low wavelengths and reached a peak of 1.6 for the undoped sample, while the 3% and 5% thin films peaked at 1.5 and 1.3 respectively. It was observed that at low photon energy, the highest value for undoped film is 2.1 while the maximum values for 3% and 5% doped thin films are 1.8 and 1.7 respectively. The refractive indices of the films ranged from 2.54 to 2.64 as the wavelength increased. The result is in accordant with the works of Horzuma et al., (2019) that used sol-gel dip coating had refractive indices values ranged from 2.05 to 3.26. But at higher photon energy, the 5% doped increased gradually to 2.4 while both the 3% and undoped thin films peaked at 2.1. This suggests that the refractive index increases with doping. This increase is due to annihilation of holes as the film surface becomes denser with increase in doping level. The 3% and undoped thin films maintained the same value due to imperfections of crystals. Furthermore, the denser material has a larger refractive index since more electric dipoles are triggered due to the electric field of the light radiation. The works of Vidhya et al., (2016) showed a calculated value of refractive index of Cu doped Titanium dioxide thin film annealed at 400 °C as 2.7.



Fig. 6. Plot of refractive index of the films

3.6. Extinction Coefficient

Figure 7 compares both undoped and $Cu:TiO_2$ to show how the extinction coefficient changes with photon energy. The 3% and undoped samples exhibit a high level of transparency as their extinction coefficient

approaches negative values at very low photon energy. However, the 5% doped sample initially increases rapidly and later decreases compared to the 3% and undoped films.



Fig. 7. Plot of Extinction coefficient of the undoped and doped TiO₂ thin films

3.7. Dielectric constant

By analyzing equations 3 and 4, we can gain significant insights into the dielectric constants of materials and their impact on refractive index, absorption, and extinction coefficient. $\epsilon_r = n^2 - k^2$

$$\epsilon_r = n^2 -$$

 $\epsilon_i = 2nk$

(3)

The refractive index is denoted by n and the extinction coefficient is denoted by k.

Real part of dielectric constant measures the attenuation of intensity of light occasioned by scattering and absorption while the imaginary part connotes the energy dissipation to the medium. Figure 8 (a & b) illustrates the real and imaginary components of the dielectric constants. An inverse relationship was observed between photon energy and both the real and imaginary parts of the dielectric constant. The imaginary section has a larger proportion of negative values than the real section.



Fig. 8. Plot of Dielectric constants [a] real and [b] imaginary of the films

3.8. Magnetic studies of TiO₂ thin films

Magnetic investigations of undoped and copper doped TiO₂ films were carried out using the VSM technique in a conducted study. The plotted magnetic characteristics curves in Fig 6 illustrate the magnetic dependence on the applied moment's magnetic field (M-H). The behavior of magnetism in undoped and copper-doped TiO₂ films shows how the magnetic field impacts the samples visually. The presence of copper doping in TiO₂ thin films demonstrates a connection between electron transfer and magnetic exchange interactions. The M-H loop reveals that both undoped TiO₂ and the doped samples exhibited ferromagnetic behavior, as depicted in Fig. The undoped sample has a saturation magnetization of around 4.18 $\times 10^{-4}$ (emu/g), a coercivity of 43.25, and a squareness value

of 0.054. When TiO_2 is doped with 3% copper, the magnetization increases to 5.13 $x10^{-4}$ (emu/g) and the squareness value increases to 0.057. When doped with 5% copper, the magnetization and squareness value increased to 6.08 $\times 10^{-4}$ (emu/g) and 0.064, respectively, indicating the doping effect. The doped samples' squareness values are less than 0.5, indicating that the inserted copper has not fully coupled with the titanium atom. The result is in agreement with the works of Ijeh et al., (2022). The low remnant values of all the samples lie between (0.0000229 - 0.0000393) emu/g. The M-H loop provides evidence of increased magnetization, potentially caused by the presence of oxygen vacancies. Table 1 displays the values of Ms, Mr, Hc, and the squareness ratio.



Fig. 7. Magnetic curves of the films.

Table 1. Wagnetization of minis				
Sample	Magnetic saturation	Remnant magnetization	Coercive	Squareness
	(M_s) (emu/g)	(M_r) (emu/g)	field $(H_c)(O_e)$	Value
Undoped	4.18x10 ⁻⁴	2.29 x10 ⁻⁵	43.25	0.054
3% Doped	5.13x10 ⁻⁴	2.93 x10 ⁻⁵	45.32	0.057
5% Doped	6.08 x10 ⁻⁴	3.93 x10 ⁻⁵	48.39	0.064

Table 1. Magnetization of films

3.9. Electrical studies of the films

To assess the resistivity of titanium dioxide, a four-point probe was utilized to measure the resistance encountered by charge carriers. The study in Figure 8 demonstrates the relationship between electrical conductivity (σ) and doping percentage in both pure and Cu-TiO₂ films at ambient temperature. The conductivity values of the 5% and 3% doped samples demonstrate a noticeable rise in comparison. Adding Cu to titanium crystal enhanced the conducting abilities of the samples without impacting their fundamental traits, as the findings imply.

Figure 8 displays the exploration of the impact of doping percentage on the electrical conductivity (σ) of undoped and Cu-TiO₂ films at ambient temperature. The 5% doped sample exhibits a significant increase in conductivity value compared to the 3% doped sample. The conductivity of the TiO₂ sample improved with the addition of Cu into the titanium crystal, as indicated by the results. It was observed that for the 3% and 5% doped films, there was evidence of sharp increase in conductivity to 0.03 Ω cm⁻¹ and 0.07 Ω cm⁻¹ respectively while the undoped film was 0% as it was devoid of carrier

density. Also the works of Ijeh et al., (2020) showed that the conductivity for copper doped MoO₃ varies from 1.0 to $1.5 \ \Omega \text{cm}^{-1}$.

The higher doping percentage results in a greater concentration of carriers, leading to an increase in electrical conductivity.



Fig. 8. Electrical conductivity versus doping of TiO₂ thin films

4. CONCLUSION

The crystallite showed only anatase phase structure and the surface morphology of the films indicated the influence of copper doping on titanium dioxide. The energy bandgap decreased slightly with increase in doping. The result of FTIR revealed functional group bonds which validated Cu-doped TiO₂. The incorporation of titanium in Cu₂O leads to thrilling advances in optical improvements. Characteristics of films of Cu₂O regarding their magnetic and electrical properties. This research work is genuine for photovoltaic applications and optoelectronic devices.

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